

## GENERIC ANALYSIS AND MORPHOLOGICAL STUDY OF TEXTURED MGO FILMS BY THE SPRAY PYROLYSIS TECHNIQUE AS BUFFER LAYERS FOR COATED CONDUCTORS

**\*Suhas M. Sabnis**

*Institute of Sciences, of J.J.T. University, Jhunjhunu, Rajasthan*

*\*Author for Correspondence*

### ABSTRACT

The prospect of using squirt pyrolysis as a uncomplicated, chemical, low cost process for the manufacturing of MgO slim films is broadly used as buffer layers intended for second generation encrusted conductors is studied. MgO films were created on Si (100) single crystal substrates and diverse exterior morphologies were pragmatic at diverse deposition temperatures. C-axis textured MgO films with soft morphologies were deposited on Si (100) single crystal. Rocking curves exposed an exceptional out of plane texture with a FWHM between 0.950 and 1.010. As the general deposition time was improved from 32 min to 64 min, the width and the rms coarseness of the produced film was just about doubled demonstrating that a good organization over the procedure was established.

**Keywords:** MGO Films, Spray Pyrolysis

### INTRODUCTION

The squirt pyrolysis system presents plentiful recompense such as ease, low cost, non toxic precursors, good reproducibility, and no need for vacuum; it has been therefore used over the years for the production of slim films of simple oxides, mixed oxides, metallic spinel type oxides, and chalcogenides films (Patil, 1999). MgO thin films have been extensively used as a chemically steady buffer layer for the deposition of high T<sub>c</sub> superconducting films and perovskite-type ferroelectric films because of its good lattice matching with mentioned materials and low chemical reactivity. Many ferroelectric and superconducting oxide films, such as PZT, LiNbO<sub>3</sub>, BaTiO<sub>3</sub> and YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7</sub> have been organized on Si substrates using MgO as a middle layer (Fork *et al.*, 1991; Desisto & Henry, 1991; Kim *et al.*, 1996; Senzaki & Mitsunaga, 1996; Yoon *et al.*, 1996). Additionally, MgO has there compense of a wide band gap, low optical loss, and a relatively low refractive index that permits confined optical modes in ferroelectric materials. Lately, MgO thin films have also been functional as shielding layers of dielectrics to improve the discharge quality and the panel's life span in an ac-plasma display panel (ac-PDP) (Kim *et al.*, 2000). A number of methods for the preparation of MgO films and coatings have been developed all through the last few years, such as pulsed laser deposition (PLD), magnetron sputtering, electron beam evaporation, metal organic chemical vapour deposition (MOCVD), and spray pyrolysis (Ishiguro *et al.*, 1996; Shih *et al.*, 1991; Talacchio *et al.*, 1989; Lu *et al.*, 1993; Yi *et al.*, 1996). Ultrasonic spray pyrolysis has recently emerged as a variable deposition process capable of developing high quality oxide and sulphide slim films. The Ultrasonic spray pyrolysis technique is based on the thermal decomposition of the source solution sprayed by an ultrasonic nozzle onto the heated substrate surface. Compared with other deposition Strategies, the ultrasonic spray pyrolysis technique possesses the benefit of simplicity, low equipment cost, high deposition rate and good thickness evenness over a large area. In this study, high quality MgO thin films have been successfully deposited by the ultrasonic spray pyrolysis, and the characteristics of obtained MgO films were studied by XRD, SEM and SRM. The growth mechanisms of MgO films deposited at different temperature were discussed. The main focus in this paper will be to produce MgO thin films, widely used as buffer layers in the coated conductor architectures.

One of the problems for the commercialisation of the coated conductors is the high cost related to the expensive physical methods employed for the fabrication of the YBCO superconducting material and the essential buffer layers (Malozemoff *et al.*, 2000). Recently low cost chemical routes such as MOD have been employed to substitute the complex physical methods (Rupick *et al.*, 2004; Paranthaman *et al.*,

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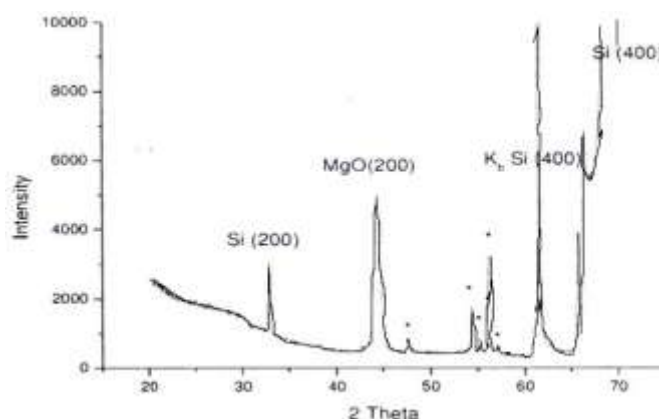
2006). Despite spray pyrolysis being one of the most well known low cost chemical deposition method for thin films, it has not been widely used for the production of either the superconducting layers or the buffer layers for coated conductor architectures. There are few reports referring to the production of YBCO films by the aforementioned method (Supard *et al.*, 2003; Ferreri *et al.*, 2001). Previously, (Shields *et al.*, 2002) has reported the production of YBCO films on STO (100) single crystal substrates (Shields & Kawano *et al.*, 2002). The precursor solutions have been prepared by dissolving  $\text{Y}_2\text{O}_3$   $\text{BaCO}_3$  and  $\text{CuO}$  in a mixture of nitric acid and distilled water. Good epitaxial films were produced with a  $J_c$  of around  $1.9 \times 10^5 \text{ A cm}^{-2}$  (77K, 0 Tesla). Furthermore, YBCO has been deposited on buffered Ni using the same spray parameters identified for the successful deposition of YBCO on STO single crystals (Shields *et al.*, 2002).  $\text{MgO}$  thin films have been widely used as buffer or template layers for YBCO coated conductor architectures (Paranthaman *et al.*, 2005; Groves *et al.*, 2001).  $\text{MgO}$  has been deposited on various substrates by spray pyrolysis but once again the texture of the obtained films is not the main focus of these studies (Bian *et al.*, 2004; Kim *et al.*, 2000; Fu *et al.*, 1999; Stryckmans *et al.*, 1999).

## MATERIALS AND METHODS

The experimental set up has been described by Pavlopoulos, 2007 briefly; the atomization of the initial chemical solution was achieved with the help of a commercial ultrasonic nebuliser. The substrate was attached to a stainless steel holder, which was positioned in the middle of the cylindrical furnace (height 30 cm, diameter 15 cm). Argon gas was used to transport the mist produced in the nebuliser onto the substrate. During spraying the temperature of the substrate was observed to drop 5-70C for each 15 sec spray. After 15 sec, therefore, the gas flow was turned off for 1 min and 45 sec and the substrate was allowed to stabilize to its set temperature. The solutions were prepared by dissolving an appropriate amount of  $\text{Mg}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  in 100 ml of distilled water for the production of  $\text{MgO}$  thin. The structural properties of the films were studied by the XRD technique and rocking curves with the aid of X Ray Diffractometer. Surface morphology and fracture cross section thickness were examined by SEM and surface roughness by contact mode AFM.

### Preparation of MgO Thin Films

In this section we will concentrate on the efforts to produce biaxially textured  $\text{MgO}$  thin films by depositing them on Si (100) single crystal substrates. In this case the lattice mismatch is 28.8 % but if the  $\text{MgO}$  unit cell rotates by  $45^\circ$  the lattice mismatch is equal to 9.7 %.

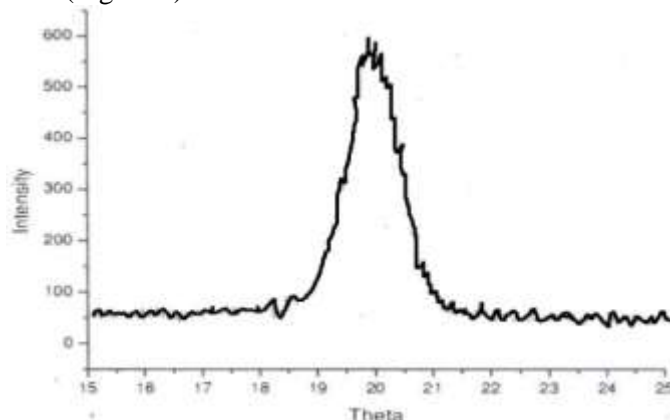


**Figure 1: XRD pattern of the  $\text{MgO}$  film deposited on Si (100) at  $650^\circ\text{C}$  from a 0.078 M solution**

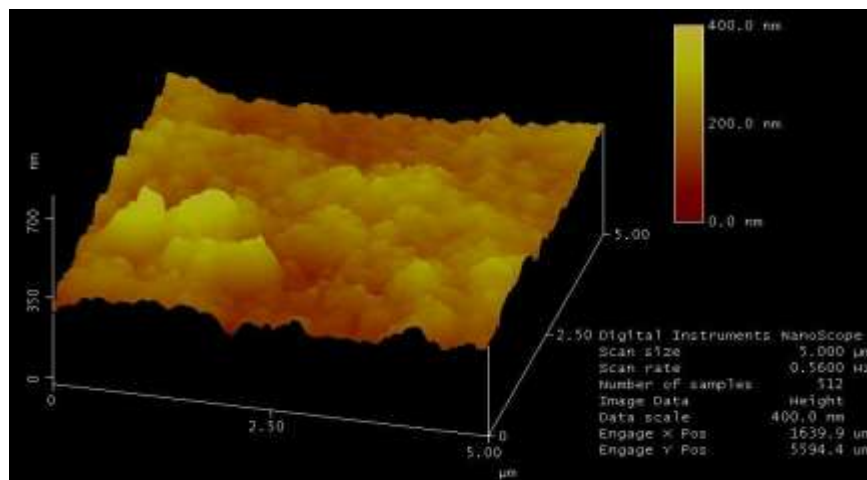
Figure 1 illustrates the XRD pattern obtained from the film deposited at  $670^\circ\text{C}$  from a 0.078 M solution- the deposition time was set to 64 min and the flow rate was fixed to 1.0l/min. The films are c-axis oriented since only the peaks from the Si substrate and the  $\text{MgO}$  (200) plane are detected. The peaks noted as \* can be attributed to high order reflections from the substrate. The rocking curve of the (200)

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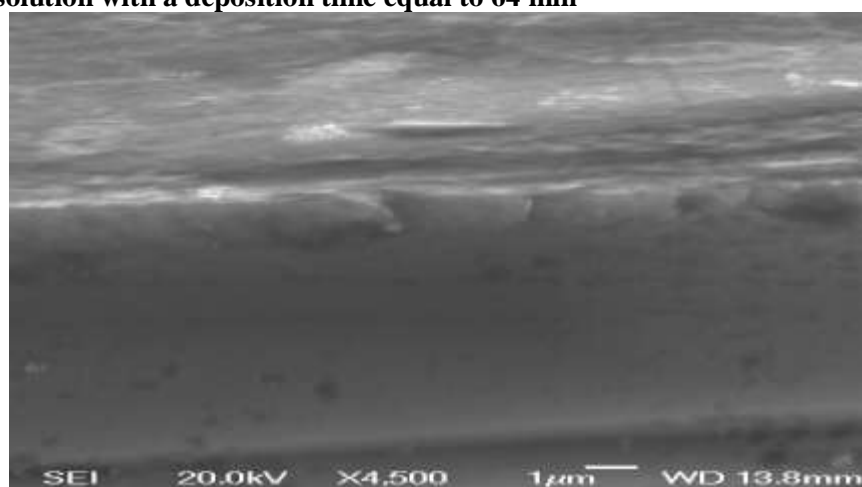
peak of the deposited film shows a FWHM equal to 0.950 indicating that the produced film presented an excellent out of plane texture (Figure 2).



**Figure 2: Rocking curve of the MgO (200) peak of the MgO film deposited on Si (100) at 670°C from a 0.078 M solution**



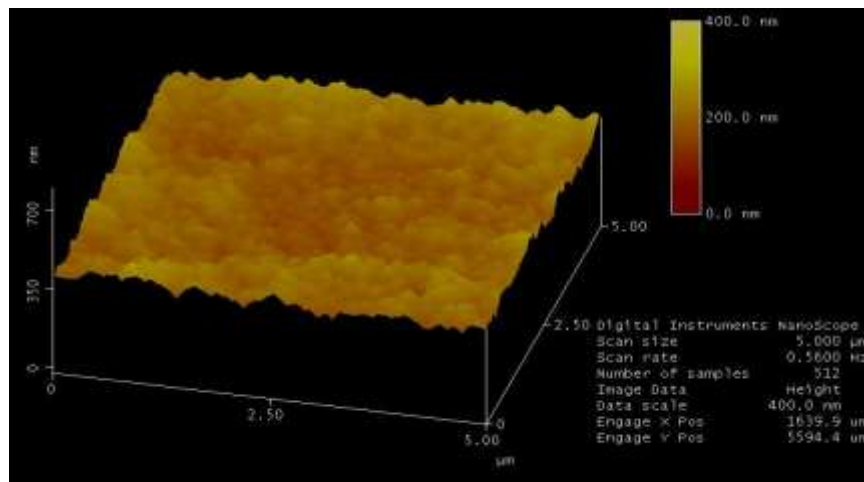
**Figure 3: 3-D 5\*5  $\mu\text{m}^2$  AFM picture of the MgO film deposited on Si (100) at 670°C from a 0.078 M solution with a deposition time equal to 64 min**



**Figure 4: SEM picture of the fracture cross section of the MgO film deposited on Si (100) at 670°C from a 0.078 M solution with a deposition time equal to 64 min**

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The rms roughness calculated from the 3-D 5\*5  $\mu\text{m}^2$  AFM picture is 43.3 nm (Figure 3). SEM studies of the cross section of the fractured surface (Figure 4) illustrate a dense and homogeneous MgO layer with film thickness estimated between 1650 to 1750 nm.



**Figure 5: 3-D 5\*5  $\mu\text{m}^2$  AFM image of the MgO film deposited on Si (100) at 650°C from a 0.078 M solution with a deposition time equal to 32 min**

Films with reduced thickness were obtained when the deposition time was reduced to 32 min temperature; solution concentration and flow rate were kept constant. The thickness of the film is between 650 to 750 nm. The produced film is once more c-axis textured as identified from the XRD pattern. The FWHM as calculated from the rocking curve is  $1.01^\circ$ . The rms roughness as calculated from the 3D AFM picture in a 5\*5  $\mu\text{m}^2$  area is 22.6 nm (Figure 5). Figure 11: 3-D 5\*5  $\mu\text{m}^2$  AFM image of the MgO film deposited on Si (100) at 650°C from a 0.078 M solution with a deposition time equal to 32 min.

## CONCLUSION

C-axis textured MgO (200) films with various deposition times were obtained on Si (100) single crystal. The typical characteristics of island growth and layer growth mechanism were observed at deposition temperatures of 650 and 680  $^\circ\text{C}$ , respectively. As the overall deposition time was increased from 32 min to 64 min, the thickness and the rms roughness of the produced film was approximately doubled indicating that a good control over the process was acquired. All the films exhibited an excellent out of plane texture with a FWHM varying from 0.950 to 1.010. The dependence of film microstructure and micrograph on substrate temperature can be explained in terms of deposition modes and growth mechanism. High quality MgO thin films may be obtained through the control of deposition modes and growth mechanism for the spray pyrolysis method. Results show that our MgO films deposited at the optimal condition (670  $^\circ\text{C}$ ) have much better quality than those in previous spray pyrolysis research. The MgO thin films with this quality should be suitable as a buffer layer for the subsequent growth of oriented ferroelectric materials.

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